# High Mobility Nanocrystalline Silicon Transistors on Clear Plastic Substrates

Alex Z. Kattamis, Russell J. Holmes, I-Chun Cheng, Ke Long, James C. Sturm, Stephen R. Forrest, and Sigurd Wagner

Abstract—We demonstrate nanocrystalline silicon (nc-Si) top-gate thin-film transistors (TFTs) on optically clear, flexible plastic foil substrates. The silicon layers were deposited by plasma-enhanced chemical vapor deposition at a substrate temperature of 150 °C. The n-channel nc-Si TFTs have saturation electron mobilities of 18 cm<sup>2</sup>V<sup>-1</sup>s<sup>-1</sup> and transconductances of 0.22  $\mu$ S $\mu$ m<sup>-1</sup>. With a channel width to length ratio of 2, these TFTs deliver up to 0.1 mA to bottom emitting electrophosphorescent organic light-emitting devices (OLEDs) which were fabricated on a separate glass substrate. These results suggest that high-current, small-area OLED driver TFTs can be made by a low-temperature process, compatible with flexible clear plastic substrates.

*Index Terms*—Clear plastic (CP) foil, electrophosphorescence, flexible substrate, nanocrystalline silicon (nc-Si), organic light emitting device (OLED), thin-film transistor (TFT).

## I. INTRODUCTION

HIN-FILM transistors (TFTs) based on nanocrystalline silicon (nc-Si) are candidates for active-matrix backplanes on clear, flexible plastic substrates. Nanocrystalline-Si TFTs qualify for CMOS [1] with electron mobilities up to  $150 \text{ cm}^2 \text{V}^{-1} \text{s}^{-1}$  [2], [3] and hole mobilities up to  $1 \text{ cm}^2 \text{V}^{-1} \text{s}^{-1}$ [4], [5]. Such nc-Si TFTs have been fabricated on glass [6], [7] and on DuPont Kapton E plastic foil [8]. Kapton E limits light transmission in the blue and therefore cannot be used with bottom emitting organic light emitting devices (OLEDs). Since nc-Si TFTs are fabricated by plasma-enhanced chemical vapor deposition at substrate temperatures from 150 °C to 300 °C, they have the potential to serve as a high-transconductance alternative to amorphous Si (a-Si) TFTs, particularly on plastic substrates requiring low process temperatures. Top gate nc-Si TFTs, however, are hampered by charge trapping in the SiO<sub>2</sub> gate insulator, leading to electrical instabilities and high threshold voltages  $(V_T)$  [4]. However, their ON currents in saturation are fairly stable making them suitable for driving OLEDs.

The objective of this work was to determine whether nc-Si TFTs with small channel width to length (W/L) ratios can serve

The authors are with the Department of Electrical Engineering and the Princeton Institute for the Science and Technology of Materials, Princeton University, Princeton, NJ 08544 USA.

Digital Object Identifier 10.1109/LED.2005.861256



Fig. 1. Optical transmission spectrum of a 50- $\mu$ m thick CP substrate. The optical transmittance of the uncoated substrate is >80% over the visible spectrum. Inset: nc-Si TFT array on a clear plastic substrate.

as OLED drivers. Therefore, we fabricated nc-Si TFTs with  $W/L = 90/45 \ \mu m$  on clear plastic substrates to drive electrophosphorescent OLEDs separately deposited on glass [9]. Due to their high mobilities, the TFTs can be integrated with bottom emitting OLEDs into display driver pixels, where the circuit covers less than 15% of the total emission aperture.

### **II. EXPERIMENTAL PROCEDURES**

We used a 50- $\mu$ m-thick clear plastic (CP) substrate with a glass transition temperature of 325 °C. The n-channel nc-Si TFTs were configured in a staggered top gate, bottom source/drain (S/D) geometry [4]. Fig. 1 shows >80% transmission for the CP substrate over the entire visible spectrum as measured using a spectrophotometer. The inset shows a  $6 \times 6$ cm<sup>2</sup> array of nc-Si TFTs. Except for the clear band along its right edge, the substrate is covered by  $SiN_x$  passivation, which gives it a yellow tint. Alloying the  $SiN_x$  with oxygen to form SiON will lead to an improved transparency [10]. The Si layers used for the TFT channel were deposited at 150 °C, and the thermally evaporated contact metals were deposited at room temperature. The Cr layers were etched using CR-7S [11], and a mixture of water, phosphoric, acetic, and nitric acids in the ratio 16:2:1:1  $H_3 PO_4 : H_2O : HNO_3 : CH_3 CO_2 H$  was used to etch Al. The SiO<sub>2</sub> gate insulator was etched using 1:10 HF:  $H_2O$ , and the Si and SiN<sub>x</sub> layers by a 4:1 CF<sub>4</sub> :  $O_2$  plasma.

Manuscript received September 12, 2005; revised October 18, 2005. This work was supported in part by the New Jersey Commission on Science and Technology and the Universal Display Corporation, and in part by a Princeton Program in Plasma Science and Technology Graduate Fellowship. The review of this letter was arranged by Editor J. Sin.

The plasma frequency for reactive-ion etching (RIE) and deposition was 13.56 MHz at a power density of  $100 \text{ mW/cm}^2$  and a pressure of 500 mtorr, except where otherwise noted.

Passivation layers of 200-nm-thick  $SiN_x$  were first deposited on both faces of the CP using a gas composition of 40:10:1  $H_2$ : NH<sub>3</sub>: SiH<sub>4</sub> at 110 mW/cm<sup>2</sup>. Next, 20-nm-thick a-Si:H followed by 40-nm-thick nominally undoped nc-Si were deposited as a channel seed layer [8]. The a-Si:H was deposited at 110 mW/cm<sup>2</sup> using 1:1 H<sub>2</sub> : SiH<sub>4</sub>. The SiH<sub>4</sub> : H<sub>2</sub> ratio was increased to 50:1 for the nc-Si deposition at 80 MHz and dichlorosilane (DCS) was added at 0.5% of the total gas flow to electrically compensate the nc-Si [12]. After thermal evaporation of the 30-nm-thick Al and 10-nm-thick Cr bottom S and D contacts, the films were etched to the CP surface forming  $100 \times 100 \ \mu m^2 \ SiN_x$  islands. The 50-nm-thick n<sup>+</sup> nc-Si S/D layer was then deposited from 120:1:1 H<sub>2</sub> : SiH<sub>4</sub> : PH<sub>3</sub> at 80 MHz. The S and D regions were patterned by RIE to expose the seed layer. Next, a 50-nm-thick nc-Si channel was deposited onto the exposed seed layer from a  $15:3:1 \text{ H}_2$  : SiH<sub>4</sub> :DCS source gas mixture at 80 MHz. This was immediately followed by deposition of a 300-nm-thick SiO<sub>2</sub> gate dielectric deposition using 350:2:1 He:  $N_2O$  : SiH<sub>4</sub> at 2 torr and 110 mW/cm<sup>2</sup>. After the S and D contact holes were etched, a 50-nm-thick Al layer was deposited by thermal evaporation. Finally, the top Al layer was patterned, defining the gate, S and D regions.

Electrophosphorescent OLEDs were separately grown on glass substrates pre-coated with a 150-nm-thick layer of indium tin oxide (ITO), degreased with solvents, and cleaned by exposure to UV-ozone following previous reports [13]. The entire device was deposited by thermal evaporation in a vacuum system with a base pressure of  $10^{-7}$ torr. The OLED consists of a 30-nm-thick 4-4'-bis[N-(1 -naphthyl-)N-phenyl-amino] biphenyl (NPD) hole transport layer deposited on the ITO anode. Next, the green phosphorescent material, fac tris(2-phenylpyridine) iridium (Ir (ppy)\_3) was doped at 8 wt.% into an N,N'-dicarbazolyl-4-4'-biphenyl (CBP) host forming the 30-nm-thick emissive layer. Finally, a 40-nm-thick electron transport layer consisting of 2,9-dimethyl-4,7-diphenyl-1,10-phenanthroline (bathocuproine, BCP) was deposited and used to confine holes and excitons within the emissive layer of the device. Cathodes consisting of a 0.5-nm-thick layer of LiF followed by a 50-nm-thick layer of Al were patterned by deposition through a shadow mask affixed in a N2 ambient and with an array of 1-mm diameter openings. The OLEDs exhibit quantum and power efficiencies of  $(7.0 \pm 0.7)\%$  and  $(20.5 \pm 2.1)$ lm/W, as is typical for  $Ir(ppy)_3/CBP$ -based devices [13].

#### III. TFT AND TFT/OLED CIRCUIT RESULTS

The characteristics of the TFTs and the OLEDs were measured using an HP4155 parameter analyzer and a calibrated photodiode [14]. The TFTs had  $V_T = 13 \pm 0.5$  V, a current ON/OFF ratio of  $I_{\rm ON}/I_{\rm OFF} > 10^8$ , a subthreshold slope of S = $1 \pm 0.1$  V/dec, and a transconductance of  $g_{\rm msat} = 0.22 \pm$  $0.03 \,\mu S \mu {\rm m}^{-1}$ , from which we extract linear and saturation electron mobilities of  $\mu_{\rm lin} = 16 \pm 2 \,{\rm cm}^2 {\rm V}^{-1} {\rm s}^{-1}$ , and  $\mu_{\rm sat} =$  $18 \pm 2 \,{\rm cm}^2 {\rm V}^{-1} {\rm s}^{-1}$ , respectively. The large  $V_T$  is due to charge



Fig. 2. (a) Drain current versus gate to source voltage characteristic of a nc-Si TFT for  $V_{\rm DS} = 10$  V. (b) Luminance versus voltage characteristics for the electrophosphorescent, green-emitting OLED and the TFT/OLED circuit. The gate to OLED cathode voltage,  $V_{\rm GC}$ , was swept from 0 to 25 V at a drain supply voltage of  $V_{\rm DD} = 15$  V. Inset: TFT/OLED circuit schematic.

trapping in the SiO<sub>2</sub>, [4] and research is underway to develop a more stable low temperature ( $\sim 150$  °C) SiO<sub>2</sub> dielectric. The nc-Si TFTs exhibit similar  $V_T$ , but slightly lower mobilities, than previous results for similar devices deposited on glass and Kapton E [4], [8]. To prevent cracking during substrate flexion, the S/D metal and n<sup>+</sup> nc-Si were thinner than in those previous devices, causing increased contact resistance and the observed decrease in mobility.

Fig. 2(a) shows the transfer characteristics for the nc-Si TFT at a S/D voltage of  $V_{\rm DS} = 10$  V. After separate evaluation of both the TFT and OLED, the OLED cathode was grounded, while the anode was connected to the TFT source. The TFT drain was biased at a power supply voltage of  $V_{\rm DD} = 15$  V [see inset, Fig. 2(b)]. Fig. 2(b) shows the luminance versus voltage across the OLED ( $V_{\rm OLED}$ ) and the gate-to-cathode voltage ( $V_{\rm GC}$ ) characteristics for the TFT/OLED circuit. As noted previously, W/L=2 was used to drive the 1-mm diameter OLED, which is ~30 times the ~  $100 \times 300 \ \mu m^2$  area of a typical



Fig. 3. Luminance versus supply voltage ( $V_{\rm DD}$ ) and gate-to-cathode voltage ( $V_{\rm GC}$ ) for the TFT/OLED circuit in Fig. 2(b).

subpixel [15]. Nevertheless, the TFT supplied sufficient current to obtain a peak luminance of 16 000 cd/m<sup>2</sup> at  $V_{\rm GC} = 30$  V and  $V_{\rm DD} = 35$  V, as shown in Fig. 3.

For bottom emitting displays, a small footprint pixel circuit is crucial. Previously, it has been shown that a five-TFT a-Si pixel circuit [16] with W/L = 50 driver TFTs can serve as stable OLED drivers. With a gate length of 10  $\mu$ m and 5  $\mu$ m allocated for gate, S and D overlaps, the area of each TFT needed in such a circuit is approximately  $10 \times 30 \ \mu$ m. Using standard interconnect widths of 10  $\mu$ m and a spacing of 10  $\mu$ m between devices allows for a total pixel area of ~ 4000  $\mu$ m<sup>2</sup>, without including the storage capacitor. This footprint leaves a light emitting aperture of >85% in a 100 × 300  $\mu$ m pixel.

#### **IV. CONCLUSION**

Nanocrystalline silicon TFTs have been fabricated on a CP foil substrate. These devices function as high current drive transistors for electrophosphorescent OLEDs. Due to the high electron mobilities characteristic of nc-Si, the TFT area can be made sufficiently small to allow for  $100 \times 300 \ \mu m$  pixels with an 85% emitting aperture. This letter illustrates the potential of nc-Si TFTs as pixel drivers in flexible, bottom emitting OLED displays.

#### ACKNOWLEDGMENT

The authors would like to thank The DuPont Company for supplying substrates.

#### REFERENCES

- Y. Chen and S. Wagner, "Inverter made of complementary P and N channel transistors using a single directly deposited microcrystalline silicon film," *Appl. Phys. Lett.*, vol. 75, pp. 1125–1127, 1999.
  C.-H. Lee, A. Sazonov, and A. Nathan, "High-mobility nanocrystalline
- [2] C.-H. Lee, A. Sazonov, and A. Nathan, "High-mobility nanocrystalline silicon thin-film transistor fabricated by plasma-enhanced chemical vapor deposition," *Appl. Phys. Lett.*, vol. 86, p. 222 106, 2005.
- [3] M. Mulato, Y. Chen, S. Wagner, and A. R. Zanatta, "Microcrystalline silicon with high electron field-effect mobility deposited at 230 °C," J. Non-Cryst. Solids, vol. 266–269, pp. 1260–1264, 2000.
- [4] I.-C. Cheng and S. Wagner, "Nanocrystalline silicon thin film transistors," *Proc. IEE–Circuits Devices Syst.*, vol. 150, no. 4, pp. 339–344, 2003.
- [5] T. Dylla, F. Finger, and E. A. Schiff, "Hole drift-mobility measurements in microcrystalline silicon," *Appl. Phys. Lett.*, vol. 87, p. 032 103, 2005.
- [6] T. Nagahara, K. Fujimoto, N. Kohno, Y. Kashiwagi, and H. Kakinoki, "In-situ chemically cleaning poly Si growth at low temperature," *Jpn. J. Appl. Phys.*, pt. 1, vol. 31, pp. 4555–4558, 1992.
- [7] J. Woo, H. Lim, and J. Jang, "Polycrystalline silicon thin film transistors deposited at low substrate temperature by remote plasma chemical vapor deposition using SiF<sub>4</sub>/H<sub>2</sub>," *Appl. Phys. Lett.*, vol. 65, pp. 1644–1647, 1994.
- [8] I.-C. Cheng and S. Wagner, "High hole and electron field effect mobilities in nanocrystalline silicon deposited at 150 °C," *Thin Solid Films*, vol. 427, pp. 56–59, 2003.
- [9] M. A. Baldo, D. F. O'Brien, Y. You, A. Shoustikov, S. Sibley, M. E. Thompson, and S. R. Forrest, "Highly efficient phosphorescent emission from organic electroluminescent devices," *Nature*, vol. 395, pp. 151–154, 1998.
- [10] A. Sugimoto, H. Ochi, S. Fujimura, A. Yoshida, T. Miyadera, and M. Tsuchida, "Flexible OLED displays using plastic substrates," *IEEE J. Sel. Topics Quantum Electron.*, vol. 10, no. 1, pp. 107–114, 2004.
- [11] Cyantek Corporation Inc., Fremont, CA.
- [12] R. Platz and S. Wagner, "Intrinsic microcrystalline silicon by plasma enhanced chemical vapor deposition from dichlorosilane," *Appl. Phys. Lett.*, vol. 73, pp. 1236–1238, 1998.
- [13] M. A. Baldo, S. Lamansky, P. E. Burrows, M. E. Thompson, and S. R. Forrest, "Very high-efficiency green organic light-emitting devices based on electrophosphorescence," *Appl. Phys. Lett.*, vol. 75, pp. 4–6, 1999.
- [14] S. R. Forrest, D. D. C. Bradley, and M. E. Thompson, "Measuring the efficiency of organic light-emitting devices," *Adv. Mater.*, vol. 15, no. 13, pp. 1043–1048, 2003.
- [15] J.-J. Lih, C.-F. Sung, M. S. Weaver, M. Hack, and J. J. Brown, "A phosphorescent active-matrix OLED display driven by amorphous silicon backplane," *Proc. SID*, vol. 4, no. 3, pp. 14–17, 2003.
- [16] P. Servati, S. Prakash, A. Nathan, and C. Py, "Amorphous silicon driver circuits for organic light-emitting diode displays," *J. Vac. Sci. Technol. A, Vac. Surf. Films*, vol. 20, pp. 1374–1378, 2002.